

# Force transduction by the microtubule-bound Dam1 ring

Jonathan W Armond

Department of Physics and MOAC Doctoral Training Centre,  
University of Warwick, Coventry, CV4 7AL, UK

Matthew S Turner\*

Department of Physics,  
University of Warwick, Coventry, CV4 7AL, UK

## Abstract

The coupling between the depolymerization of microtubules (MTs) and the motion of the Dam1 ring complex is now thought to play an important role in the generation of forces during mitosis. Our current understanding of this motion is based on a number of detailed computational models. Although these models realize possible mechanisms for force transduction, they can be extended by variation of any of a large number of poorly measured parameters and there is no clear strategy for determining how they might be distinguished experimentally. Here we seek to identify and analyze two distinct mechanisms present in the computational models. In the first the splayed protofilaments at the end of the depolymerizing MT physically prevent the Dam1 ring from falling off the end, in the other an attractive binding secures the ring to the microtubule. Based on this analysis, we discuss how to distinguish between competing models that seek to explain how the Dam1 ring stays on the MT. We propose novel experimental approaches that could resolve these models for the first time, either by changing the diffusion constant of the Dam1 ring (e.g., by tethering a long polymer to it) or by using a time varying load.

*Key words: Brownian ratchet; burnt bridges; DASH; protofilaments; mitosis; motors*

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\*Corresponding author. Address: Department of Physics, University of Warwick, Gibbet Hill Road, Coventry, CV4 7AL, UK, m.s.turner@warwick.ac.uk

## Introduction

Mitosis is the mechanism of cell division in eukaryotic cells. In mitosis, chromosomes condense and are arranged at the center of the cell by the mitotic spindle. Microtubules (MTs) are protein fibers, composed of  $n$  parallel protofilaments (PFs, typically  $n = 13$ ) forming a hollow cylinder. Each PF is built from stacked tubulin protein dimers. MTs emanate from centrosomes and attach to chromosome-bound kinetochores. Centrosomes are positioned at both poles of the cell forming a bipolar spindle. During anaphase chromosomes are segregated and transported to the cell poles by the retraction of MTs, providing both daughter cells with a single copy of the cell's chromosomes (1). To achieve segregation, depolymerizing kinetochore-attached microtubules (KMTs) must generate forces, e.g., to overcome chromosomal drag in the cytosol (2). There is evidence that mitotic MT force generation occurs in the absence of MT minus-end directed motor proteins (3) and when minus-end depolymerization is inhibited (4). Previously, a hypothetical sleeve had been proposed to couple MT depolymerization to kinetochores (5, 6). A 10-protein complex, purified from budding yeast (7), called Dam1 (or DASH) has been observed to form rings around MTs (8, 9). Dam1 rings have been observed tracking depolymerizing MT plus ends *in vitro* (10) and an optical trap has been used to measure force-distance traces for Dam1-coated polystyrene beads attached to depolymerizing MTs (11–13). Intriguingly, Dam1 has been shown to be essential for chromosome segregation in budding yeast (14, 15) and important for avoiding mis-segregation problems in fission yeast (16).

## Mechanisms of force transduction

Several models have been proposed to explain how the Dam1 ring can couple the kinetochore to a depolymerizing MT so as to produce a force. Over two decades ago Hill proposed the first quantitative model describing how a depolymerizing microtubule could be harnessed for the production of force (6). In this model a hypothetical sleeve surrounds the MT and provides the attachment to the kinetochore. An attraction between the sleeve and MT provides an energy barrier preventing detachment, but this sleeve may still be able to slide along the MT without paying the energy of detachment. More recent computational models are detailed, mechanistic and micromechanical. One such model has taken into account the energy predicted to be available due to the curling of PFs (17–19) and, following the discovery of the Dam1 ring, was extended to reflect current structural knowledge and incorporate the hypothesis that Dam1 forms rigid transient links to the MT (20). Another independent model postulated an electrostatic attraction maintaining the rings position at the tip of the MT (21), combined with a powerstroke. All recent models include a combination of the following features: 1) the intrinsic diffusion of the Dam1 ring; 2) an effective powerstroke due to curling PFs; 3) an attractive potential between Dam1 and the MT. However, whilst these models include many of the relevant physical features of the system and produce satisfactory simulations of a reliable force transduction system, the problem cannot be considered “solved” because many variants on these models are possible and they have not been quantitatively compared to data. Furthermore, the lack of discriminatory experimental data precludes validation. In light of this we feel that much can be gained from rigorously analyzing the contribution of the various features in order to determine their possible role.

In what follows we describe two distinct “minimal” models, both of which describe a functional Dam1-mediated force transduction system. In the *protofilament* model the splaying PFs at the depolymerizing end physically prevent the ring from sliding off. In *binding* models an attraction between the ring and MT provides an energy barrier preventing detachment. The two models are not mutually exclusive – a hybrid model, incorporating both contributions, may also apply although one of the constituent mechanisms will typically dominate. While it is straightforward to modify our analysis to include such hybrid models we neglect them here for clarity, since our purpose is to differentiate the contributions. In common with previous models we also neglect other molecular components, e.g., microtubule-associated proteins (MAPs) and kinases (22, 23), that certainly play important additional roles *in vivo*.

Some previous studies have incorporated a powerstroke, arising from the motion of PFs, in driving the motion of the Dam1 ring (17, 20, 21). Indeed, it has been demonstrated that PFs can push a bead attached to

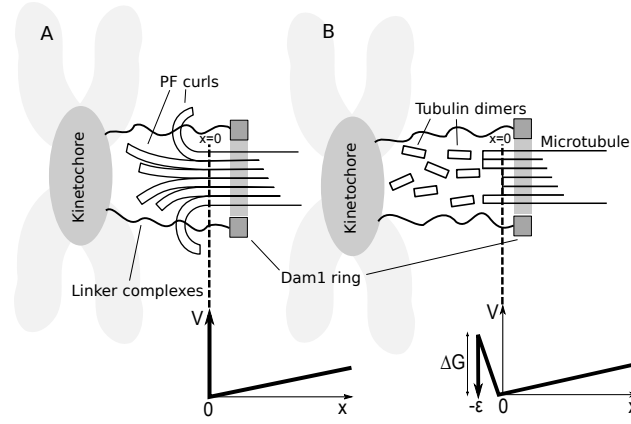


Figure 1: Two general classes of models of Dam1 ring-microtubule coupling. In both cases force is generated by rectifying Brownian motion, that is the ring diffuses to the right and the MT happens to unzip one segment or it is driven to the right by a powerstroke associated with unzipping. The dotted line denotes the point reached by MT unzipping ( $x = 0$ ). (A) The ring is sterically confined to the MT by PFs (protofilament model) (B) The ring is attractively bound to the MT surface with a free energy of binding  $\Delta G_{\text{Dam1}}$  (binding model). Below each model, the potential profile  $V$  in which the ring diffuses is shown as a function of the distance  $x$  of the ring from the MT end, and a dotted line indicates the connection between profile and model. The load force is the slope of  $V(x)$  for  $x > 0$ . In (A) there is a large (infinite) energy barrier preventing the Dam1 ring moving to  $x < 0$  whenever curled PFs are present. If the PFs completely depolymerize, leaving a ‘blunt’ end on the MT, this barrier disappears. In (B) the ring maintains only partial contact as it slides off the end of the MT ( $-\epsilon < x < 0$ ), which results in a rise in energy until it finally loses contact and is lost forever for  $x < -\epsilon$ . See text for details.

the side of a MT (24), with a force of about 5 pN per 1–2 PFs. However, it is also known that models in the “burnt bridges” (25) class require no powerstroke *per se* to generate motion (26). Rather, purely diffusive Brownian motion can be rectified if “bridges” (here segments of MT) are lost (depolymerize) after they have been crossed. No instantaneous *physical* force is required, although the resultant rectified Brownian motion does give rise to a force in the thermodynamic sense. Such models are, in turn, members of a larger class of models known as “Brownian ratchets” (27). These models exhibit velocities that depend on applied force, and stall for sufficiently high forces, as also seen in more complex models (21). It is not clear *a priori* to what extent the powerstroke plays an important role. In the present work we also seek to answer this question.

## Generalized model

We seek to analyze a general model that includes both a diffusive burnt bridge mechanism and a powerstroke, in order to determine their relative contribution. Here the powerstroke involves a depolymerization event which “unzips” PFs and moves the position of the last unbroken section of MT; a new section takes on this identity when the previous one unzips (contains separated, splayed protofilaments). As a highly energetic powerstroke this is assumed to occur even when the ring is very close to the MT end, with a rate that gives rise to a depolymerization velocity  $v_{\text{ps}}$ . The sequence of microscopic PF unzipping events gives rise to a well defined velocity for the last fully intact MT section, irrespective of the sequence in which the neighboring PFs unzip, and the precise MT helicity. Critically, we also assume that polymerized MT can be lost with a *second rate*, giving a depolymerization velocity  $v_{\text{bb}}$ , whenever the ring has diffused a distance  $\delta$  from the end. This can be thought of as the depolymerization velocity of a bare MT because, in this case, there is no Dam1 ring anywhere on the MT. We make no prior assumptions as to which contribution dominates, rather we determine this by fitting the parameters  $v_{\text{bb}}$ ,  $v_{\text{ps}}$  and  $\delta$  to data for the variation of the Dam1 velocity with load (12).

Our model involves a clear distinction between two mechanisms (only) and represents the simplest possible model capable of explaining this data. It can be biophysically motivated on the grounds that the Dam1

ring interacts with neighboring tubulin and so the rate of PF unzipping at the MT end should depend on how close the Dam1 ring is to the end (a concept already introduced in (21)). In the section below we discuss this mechanism in terms of a putative energy landscape for the depolymerization (unzipping) reaction. We believe that it would be unjustified to postulate the existence of any features on this energy landscape beyond the minimum required to explain the data. This amounts to a model involving two (distinct) depolymerization mechanisms. Our results suggest that both mechanisms play important roles at moderate loads. In particular powerstroke-only models are clearly inconsistent with the data showing a velocity that is dependent on force (11, 12), assuming a strong power-stroke as previously measured on the order of 30 – 65 pN (24). This is because, with a powerstroke-only model, we would not expect the velocity of Dam1 ring to be significantly slowed under a force as low as 2 pN; the data shows a significant slowing. We find that the length scale  $\delta$  controlling burnt-bridge reactions, a free fit parameter, is close to the axial length of a tubulin dimer. We speculate that this may provide indication of “crack”-like splitting of the MT, as discussed below.

## Model

The Dam1 ring complex is reported to be capable of axial movement with respect to the MT (10). Therefore, we treat the Dam1 ring as a particle undergoing one-dimensional Brownian motion in a potential  $V(x)$  (shown for two different models in Fig. 1). The fully intact MT extends away from the depolymerizing end for  $x > 0$  and the point at which the MT lattice unravels is  $x = 0$  (see Fig. 1 A). The following Fokker-Plank equation (28) determines the probability density  $\phi(x, t)$  for the ring’s position relative to the (moving) end,

$$\frac{\partial \phi}{\partial t} = D \frac{\partial}{\partial x} \left( \frac{\partial \phi}{\partial x} + \frac{1}{k_B T} \frac{\partial V}{\partial x} \phi \right), \quad (1)$$

where  $D$  is the diffusion constant of the ring. This approach is appropriate providing the depolymerization velocity  $v$  of the MT is not too fast (bounds given later in this section), otherwise we must instead treat this as a full moving boundary problem. Since the microtubule depolymerization is here quasistatically slow with respect to the diffusive relaxation of the ring, we can neglect the drag force on the ring, except as discussed in Supporting Material.

In the following we assume the Dam1 ring is sufficiently stable that it can only dissociate by slipping off the tip. For simplicity we restrict our analysis to continuous depolymerization processes only and discount the possibility of rescue and polymerization. Although it would be straightforward to include such processes we believe that they would distract from the central results of this paper.

A force  $-\partial V/\partial x$  appears in Eq. 1. This is the magnitude of the applied force  $f$  on the Dam1 ring while on the MT ( $x > 0$ ) since the ring must do work to move against this force. Hence, from Eq. 1 it can be shown that, for constant (or slowly varying)  $f$  the probability distribution  $\phi(x)$  is of Boltzmann form

$$\phi(x) = \frac{f}{k_B T} \exp \left( -\frac{fx}{k_B T} \right), \quad (2)$$

where the ring typically explores a characteristic diffusion length  $\lambda = k_B T/f$  from the MT end and positive values of  $f$  here indicate loads pulling in the negative  $x$  direction (towards the MT end). We assume that the depolymerization is *quasistatically slow*. This is appropriate provided the time for the MT to depolymerize the distance  $\lambda$  is much larger than the relaxation time for a ring to diffuse this distance. This in turn requires  $\lambda/v(f) \gg \lambda^2/D$ . In this case the distribution of the ring position is always close to the equilibrium probability distribution that it would have on an MT that was not depolymerizing. This sets an upper bound on the depolymerization velocity, or equivalently a lower bound on the load force, beyond which our theory is at best semiquantitative; solving  $D/v(f) = k_B T/f$  with Eq. 4, we estimate these values to be 500 nm/s and 0.04 pN respectively. Under these conditions the average ring velocity is equivalent to the depolymerization velocity of the MT.

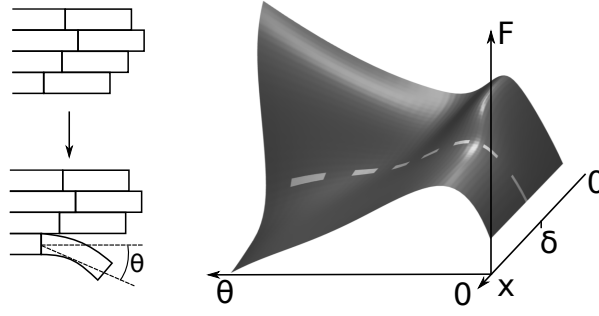


Figure 2: Schematic energy landscape underlying PF unzipping. The proposed free energy  $F$  landscape of a tubulin dimer at the end of the MT is shown (right) as a function of the distance of the Dam1 ring from the MT end,  $x$ , and a reaction coordinate for the unzipping, the angle  $\theta$  moved by the tubulin dimer (see diagram at left). The diagram is shown for illustrative purposes only and is not quantified in this work. Here  $\theta = 0$  represents a dimer in a linear PF incorporated into a stable MT. During unzipping  $\theta$  increases and the dimer moves out, ultimately forming the base of a splayed PF. The unzipping is an activated process with an energy barrier (the height of the ridge on the right) that is *different* for a powerstroke ( $x < \delta$ ) and a burnt bridges reaction ( $x > \delta$ ), leading to velocities  $v_{ps}$  and  $v_{bb}$  respectively. The energy landscape must have at least these basic features in order to give rise to the two depolymerization rates consistent with the data.

### Force dependent depolymerization velocity

The powerstroke and burnt-bridge reactions can be thought of as arising from transitions over an energy barrier of the form shown in Fig. 2, where the free energy  $F$  of PF curling is shown as varying with protofilament angle  $\theta$  and the distance of the Dam1 ring from the MT tip  $x$ . The figure shows only a putative schematic of the free energy of PF curling reaction, and should not be confused with the potential  $V(x)$  in which the Dam1 ring diffuses.

PFs may produce a power stroke that pushes the ring with force  $f_{pf}$ , estimated from experimental evidence to be 30 – 65 pN (24). This is the slope down the descending valley, diagonally right to left, in Fig. 2. Provided that the load force  $f \ll f_{pf}$  the powerstroke will give rise to a depolymerization velocity  $v_{ps}$  that is the rate at which the last intact dimer on the MT crosses the highest part of the ridge-like energy barrier in Fig. 2 ( $x < \delta$ ). Since the estimate for  $f_{pf}$  is so much larger than any force considered here, it is reasonable to make the limited assumption that  $v_{ps}$  is constant for all experimentally measurable load forces of a few pN or less.

In addition the MT can also depolymerize when the Dam1 ring is further than a critical distance  $\delta$  from the end of the MT. In this case the burnt-bridge reaction gives rise to a depolymerization velocity  $v_{bb}$  that is the rate at which the last intact dimer on the MT crosses the *lower* part of the ridge-like energy barrier in Fig. 2 ( $x > \delta$ ). That the rate of MT unzipping is *retarded* when the Dam1 ring is near the MT end is a result of the fact that the velocity decreases as the load force is increased and the ring is more often closer to the MT end. Although it is not *necessary* to interpret our model in terms of the Dam1 ring physically occluding the unzipping of the tubulin dimers, this interpretation may not be unreasonable, particularly in view of the fact that we find  $\delta$  to be comparable with the axial length of the last intact ring of tubulin dimers.

The resultant velocity due to both mechanisms is the sum of the probability that the ring is close to the MT end  $x < \delta$ , multiplied by the powerstroke velocity, and the probability that it is far  $x > \delta$ , multiplied by the burnt-bridge velocity,

$$\begin{aligned}
 v &= v_{ps} \left( 1 - \int_{\delta}^{\infty} \phi(x) dx \right) + v_{bb} \int_{\delta}^{\infty} \phi(x) dx \\
 &= (v_{bb} - v_{ps}) \int_{\delta}^{\infty} \phi(x) dx + v_{ps}
 \end{aligned} \tag{3}$$

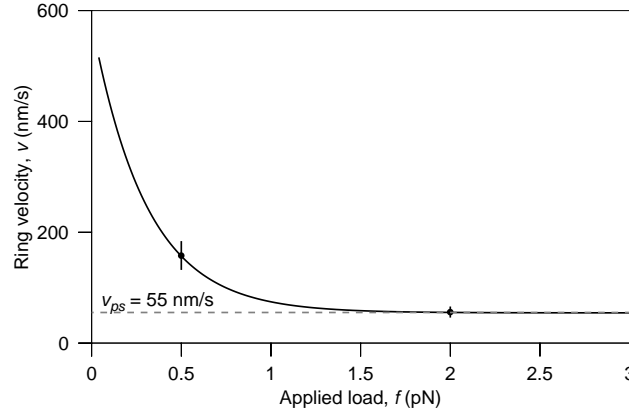


Figure 3: The variation of velocity of the Dam1 ring with applied load. The velocity falls as the force increases because the motion must increasingly rely on the energetic powerstroke. Note that, although the graph appears to suggest an absence of a stalling force, a significantly higher forces the assumption of constant  $v_{ps}$  would fail and the ring would stall. The curve is produced from the best fit of  $\delta$  and  $v_{ps}$  in Eq. 4 and data from (12).

The velocity follows from Eqs. 3 and 2

$$v = (v_{bb} - v_{ps}) \exp\left(\frac{-f\delta}{k_B T}\right) + v_{ps}, \quad (4)$$

The variation of this velocity with load is shown in Fig. 3 for  $v_{bb} = 580$  nm/s (29), and the values  $v_{ps} = 55$  nm/s and  $\delta = 14$  nm that correspond to the best fit to data (12). Since a “burnt-bridges”-only model fails to fit the data sufficiently (i.e.  $v_{ps} > 0$ ) it suggests that a powerstroke plays a role in forced Dam1 motion. It should be noted that, although in this model  $v \rightarrow v_{ps}$  as  $f \rightarrow \infty$ , we do not suggest this is a physical feature of the system. Rather it is the consequence of the assumption that protofilaments are perfectly rigid and the powerstroke reaction is asymptotically strong. Our model would need modification for forces approaching  $f_{pf}$ . As discussed later, PFs are estimated to require tens of pN to bend.

## Two models for Dam1 ring retention

We now proceed to calculate the mean time the Dam1 ring will remain on a MT and transduce force\*. This time is controlled by different physics in the protofilament and the binding models, see Figs. 1 and 4. However, in both cases, the *velocity* of the ring is governed by the model described above, see Eq. 4.

### Runtime: Binding model

The binding model involves a ring diffusing on a MT according to Eq. 1, leading to a depolymerization velocity as given in Eq. 4. However, in order to detach from the MT end the ring must overcome a linear potential imposed by the Dam1-MT binding energy  $\Delta G_{\text{Dam1}}$  as it slides off the end of the ring. In this respect it is similar to Hill’s model (6). Previous models invoked a  $\Delta G_{\text{Dam1}}$  that also determined the roughness of the energy landscape through “linkers” (20) whose existence is supported by binding studies (33). Here we don’t make this assumption, rather  $\Delta G_{\text{Dam1}}$  could be due to less specific interactions without significant energy barriers between neighboring sites (34) but, importantly, can vary independently of the diffusion

\*Recently it has been discovered that the Dam1 oligomers track the tip of depolymerizing MTs without forming a ring (30, 31). It seems unlikely that a protofilament model could operate without a full ring, however, it is not known to what extent, if at all, small oligomers contribute to force production. Furthermore, it has been shown that 16-20 Dam1 complexes are present at the kinetochore during metaphase (32), enough to form the ring. We await the result of experiments where tension is applied to putative Dam1 oligomers.

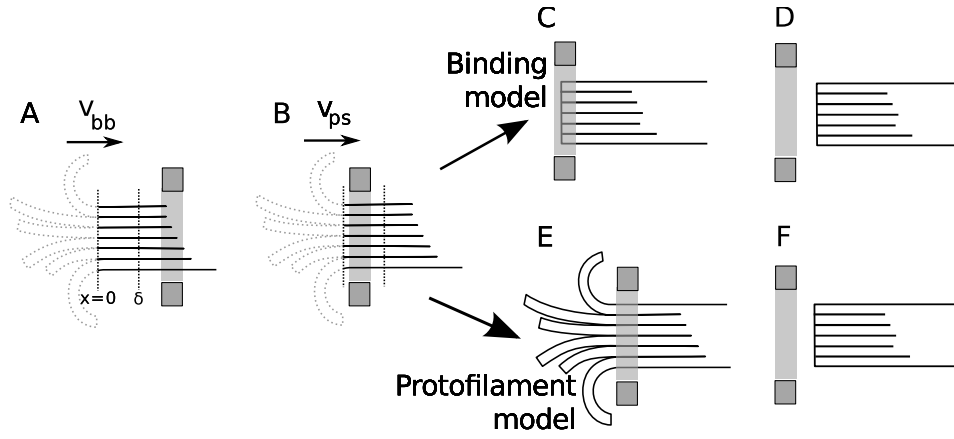


Figure 4: Various sketches of a ring on a microtubule. (A) In this configuration the ring is further than  $\delta$  from the tip of the MT, so the MT depolymerizes with velocity  $v_{bb}$ . Unzipped protofilaments are shown dotted as they do not affect depolymerization. (B) In some other configuration, the ring is closer to the tip than  $\delta$ , so the MT depolymerizes with velocity  $v_{ps}$ . In (C-F) the detachment mechanisms are shown. This is either insensitive to PFs (C-D; binding model) or sensitive to PFs (E-F; protofilament model). In (C) and (E) the ring has not yet escaped. In (D) and (F) the ring has escaped from the MT.

constant  $D$ . This, in turn, is fixed by the smoothness of the underlying energy landscape experienced by the ring as it diffuses along the MT (distinct from the energy landscape experienced by an unzipping PF shown in Fig. 2). This model assumes that the splayed PFs play no role, either because they are transient (rapidly breaking) or otherwise interact negligibly with the ring as it slides off the end of the MT. Although clearly an extreme approximation it forms the natural opposite limit to the protofilament model discussed in the next section. Under a load force the ring is in the well of a “tick”-shaped potential with two linear domains (inset Fig. 1 B). To move to the left (towards negative  $x$ ) it must partially unbind from the MT, to move to the right (positive  $x$ ) it must do work against the applied force. The potential gradients experienced by the Dam1 ring determine the load force  $f$  (while on the MT,  $x > 0$ ) and the resultant force  $f_\epsilon$  (while detaching from the MT over the small distance  $-\epsilon < x < 0$ ). The force on the ring, adopting a sign convention where a positive force acts in the direction of positive  $x$ , is therefore given by

$$-\frac{\partial V}{\partial x} = \begin{cases} -f & x \geq 0 \\ f_\epsilon = \frac{\Delta G_{\text{Dam1}}}{\epsilon} - f & -\epsilon \leq x \leq 0 \end{cases} \quad (5)$$

where  $\epsilon$  is the unbinding region. If the ring is in the region  $x < -\epsilon$  then it is lost, and if lost we assume it never returns, hence we have  $V \rightarrow -\infty$  for  $x < -\epsilon$ .

Symmetry from electron microscopy (10) and copy number (32) experiments suggest 16 complexes are required to form the Dam1 ring, however, the total bond energy may not be additive and this should therefore be regarded as an extreme upper bound on the total binding energy.

The detachment of the ring can be cast as a classical Kramers escape problem (35). To solve Eq. 1 with Eq. 5 we followed the method in (36, 37). In this way we obtain the lifetime of the metastable state directly from the Laplace transformed version of Eq. 1, with initial condition  $\phi(x, 0) = \delta(x)$ , where  $\delta(x)$  is the Dirac delta function, although the precise form of this initial condition is unimportant. The mean time the ring remains on the MT is the runtime  $\tau$

$$\tau = \frac{(k_B T)^2}{D f_\epsilon} \left( \frac{e^{\frac{f_\epsilon \epsilon}{k_B T}} - 1}{f} - \frac{e^{\frac{f_\epsilon \epsilon}{k_B T}} + \frac{f_\epsilon \epsilon}{k_B T} - 1}{f_\epsilon} \right), \quad (6)$$

where  $\epsilon$  is a small distance.

### Runtime: Protofilament model

The protofilament model involves a ring diffusing on a MT according to Eq. 1, leading to a depolymerization velocity Eq. 4, as before. However, in order to detach from the MT end the ring has to wait until all protofilaments have broken (depolymerized), leaving a sufficiently “blunt” end to the MT for the ring to simply slide off, see Fig. 1 A. We no longer require the Dam1 ring to overcome a Dam1-MT binding energy. Electron microscopy reveals that short, separated PFs splay outwards at the depolymerizing MT end (18, 19) and it is quite plausible that these block the escape of the ring; the elastic energy required to straighten a curled PF (38) follows from measurements of their rigidity (39, 40) and is of the order of tens of  $k_B T$  per subunit, i.e. very large.

The “frayed” PFs near the end of the MT are curved and laterally separate. The unzipping (depolymerization) of the MT lattice (see Fig. 1A) is most accurately described as a process which transfers length from the polymerized MT into separated PFs. The unzipping is thought to be driven by the stored elastic energy in the  $\alpha\beta$ -tubulin units in the lattice (41). When not constrained by lateral bonds, PFs relax into a curved state. We model unzipping as a Poisson process with rate  $k_{\text{unzip}}$ . Each unzipping event extends every PF curl by some microscopic, or subunit, length  $b$ , leading to a depolymerization velocity  $v = bk_{\text{unzip}}$ . This microscopic length might be the tubulin dimer repeat distance  $b$ , if the MT splits between a particular pair of PFs, or otherwise smaller than this. When the ring is within a small length  $\delta$  unzipping is inhibited. To more carefully analyze this process note that the time between unzipping events  $t_{\text{unzip}}$  is an exponential random variable, with probability density function  $p_{\text{unzip}}(t) = k_{\text{unzip}} \exp(-k_{\text{unzip}}t)$ , and mean

$$\langle t_{\text{unzip}} \rangle = \frac{1}{k_{\text{unzip}}} = \frac{b}{v}. \quad (7)$$

The distribution of the ring position in this model follows Eq. 2. Detachment occurs when all PF curl lengths reach zero<sup>†</sup>. Since  $v$  is a function of the applied force  $f$ , according to Eq. 4,  $\langle t_{\text{unzip}} \rangle$  increases under load. From Eq. 2, we have that the characteristic distance of the ring from the tip is  $\lambda = k_B T / f$ . The characteristic time for the ring to diffuse this length and escape is  $\lambda^2 / D$  which is much less than  $\langle t_{\text{unzip}} \rangle$  for typical parameters whenever  $f > 0.15$  pN (see *Supporting Material*). Thus it is not unreasonable to assume that the ring might disengage from the MT extremely rapidly as soon as all curled PFs reach zero length.

We assume that tubulin subunits on the frayed PFs break independently according to a Poisson process with rate  $k_{\text{break}}$ . The depolymerization of PFs then follows from the loss of all PF material beyond the break, as in previous computational models (42). A PF curl reaches zero length if the axial bond nearest to the unzipping point breaks, see Fig. 1 A. Since this occurs with a rate  $k_{\text{break}}$  the waiting time  $t_{\text{pf},i}$  for PF curl  $i$  to break off completely is an exponential random variable. The wait time for all  $n$  PF curls breaking is the order statistic  $t_{\text{pf}} = \max_i t_{\text{pf},i}$ . The distribution function for this time is  $P_{\text{pf}}(t) = (1 - \exp(-k_{\text{break}}t))^n$  and the mean wait time (see section 4.6 from (43) or (44)) is

$$\langle t_{\text{pf}} \rangle = \sum_{i=1}^n \frac{\langle t_{\text{pf},i} \rangle}{n-i+1} = \frac{H_n}{k_{\text{break}}}, \quad (8)$$

where  $H_n = \sum_{i=1}^n i^{-1}$  is the harmonic number, roughly  $\log n$  for  $n \gg 1$  as can be seen by converting the sum to an integral,  $\langle \cdot \rangle$  denotes the ensemble average. The Dam1 ring will therefore no longer be secured to the MT end and will detach after a time  $t_{\text{pf}}$  provided that no unzipping events having taken place during the time  $t_{\text{pf}}$ . If the MT has unzipped then the PFs extend (from their base), effectively “restarting” the waiting process.

Fundamentally we are interested in the mean runtime  $\tau$ , this being the time taken for the curled PFs to all depolymerize completely even while the MT is simultaneously undergoing stochastic unzipping events.  $\tau$

<sup>†</sup>Extensions of our model to the case of loosely-fitting rings is straightforward, involving attachment whenever the PF curls exceed some finite length  $L$ . Our results are qualitatively insensitive to this modification, provided the ring rarely detaches at low force. Furthermore, for such rings the molecular length  $b$  becomes irrelevant as the characteristic timescale is  $L/v$ .

can be found by counting the number of unzipping events  $N$  that occur before the PFs all successfully break and the Dam1 ring can disengage.  $N$  is geometrically distributed with mean  $\langle N \rangle = 1/P_{\text{detach}}$  with  $P_{\text{detach}}$  the probability that the curled PFs depolymerize completely before the next unzipping event. Thus

$$\tau = \frac{\langle t_{\text{unzip}} \rangle}{P_{\text{detach}}}. \quad (9)$$

The ring detaches if the PFs break before an unzipping occurs, i.e. with probability that  $t_{\text{pf}} < t_{\text{unzip}}$ ,

$$P_{\text{detach}} = \int_0^\infty dt p_{\text{unzip}}(t) \int_0^t p_{\text{pf}}(t') dt', \quad (10)$$

where  $p_{\text{pf}} = dP_{\text{pf}}/dt$  is the probability density function for  $t_{\text{pf}}$ . Evaluating the integral with respect to  $t'$

$$\begin{aligned} P_{\text{detach}} &= \int_0^\infty P_{\text{pf}}(t) p_{\text{unzip}}(t) dt \\ &= \int_0^\infty \left(1 - e^{-k_{\text{break}} t}\right)^n k_{\text{unzip}} e^{-k_{\text{unzip}} t} dt, \end{aligned} \quad (11)$$

Binomially expanding the integrand, integrating term-by-term and substituting back into Eq. 9 we obtain

$$\tau = \frac{1}{k_{\text{unzip}}} \left( \sum_{j=0}^n \binom{n}{j} \frac{k_{\text{unzip}} (-1)^j}{j k_{\text{break}} + k_{\text{unzip}}} \right)^{-1}. \quad (12)$$

where  $\binom{n}{j} = n!/j!(n-j)!$ .

## Time varying applied forces

We now consider an oscillating applied force of the form

$$f(t) = f_0 \sin \omega t + f_1 \quad (13)$$

Provided the period is sufficiently long  $\omega^{-1} \gg \lambda^2/D$  our quasi-static approximation for the ring position should give an accurate estimate for its probability density  $\phi(x, t)$ .

The depolymerization velocity will be retarded according to Eq. 4, relating  $v$  to  $f(t)$ .

## Protofilament model under oscillating force

The probability that the MT does *not* unzip in a time  $t$  after the time at which the last unzipping occurred  $\tilde{t}$  is

$$\bar{P}_{\text{unzip}}(t; \tilde{t}) = 1 - P_{\text{unzip}}(t; \tilde{t}) = \exp \left( - \int_{\tilde{t}}^{\tilde{t}+t} \frac{v(t')}{b} dt' \right), \quad (14)$$

Since Eq. 14 depends explicitly on  $\tilde{t}$ , we perform an average over  $\tilde{t}$ , appropriately weighted, to give the complementary distribution of times between unzipping events

$$\begin{aligned} \bar{P}_{\text{unzip}}(t) &= \int_0^{2\pi/\omega} \bar{P}_{\text{unzip}}(t; \tilde{t}) \frac{v(\tilde{t})}{\mathcal{N}b} d\tilde{t} \\ &= \int_0^{2\pi/\omega} \exp \left( - \int_{\tilde{t}}^{\tilde{t}+t} \frac{v(t')}{b} dt' \right) \frac{v(\tilde{t})}{\mathcal{N}b} d\tilde{t} \end{aligned} \quad (15)$$

involving a normalization constant  $\mathcal{N} = \int_0^{2\pi/\omega} \frac{v(\tilde{t})}{b} d\tilde{t}$ .

To calculate the runtime as in Eq. 9, we first determine the probability the unzip time exceeds the curled PF breaking time

$$\begin{aligned}
 P_{\text{detach}} &= \int_0^\infty dt p_{\text{unzip}}(t) \int_0^t p_{\text{pf}}(t') dt' \\
 &= \int_0^\infty dt' p_{\text{pf}}(t') \int_{t'}^\infty p_{\text{unzip}}(t) dt \\
 &= \int_0^\infty \bar{P}_{\text{unzip}}(t) p_{\text{pf}}(t) dt.
 \end{aligned} \tag{16}$$

The probability density of  $t_{\text{pf}}$  is

$$\begin{aligned}
 p_{\text{pf}}(t) &= \frac{d}{dt} P_{\text{pf}}(t) = \frac{d}{dt} \left( 1 - e^{-k_{\text{break}} t} \right)^n \\
 &= n k_{\text{break}} e^{-k_{\text{break}} t} \left( 1 - e^{-k_{\text{break}} t} \right)^{n-1}.
 \end{aligned} \tag{17}$$

Finally the runtime is

$$\begin{aligned}
 \tau &= [P_{\text{detach}} \langle k_{\text{unzip}}(t) \rangle]^{-1} \\
 &= \frac{1}{P_{\text{detach}}} \frac{\omega}{2\pi} \int_0^{2\pi/\omega} \frac{b}{v(t)} dt,
 \end{aligned} \tag{18}$$

where  $1/P_{\text{detach}}$  is the mean number of steps before detachment.

### Binding model under oscillating force

The generalization of Eq. 6 to the case of time-varying force (Eq. 13) is straightforward,

$$\tau = \frac{\omega}{2\pi} \int_0^{2\pi/\omega} \frac{(k_B T)^2}{D f_\varepsilon} \left( \frac{e^{f_\varepsilon \varepsilon / k_B T} - 1}{f} - \frac{e^{f_\varepsilon \varepsilon / k_B T} + f_\varepsilon \varepsilon / k_B T - 1}{f_\varepsilon} \right) dt, \tag{19}$$

where  $f$  and  $f_\varepsilon$  are now *time-dependent* potential gradients, according to Eq. 13 with Eq. 5.

## Results and Discussion

We identified the following parameters using data reported in the experimental literature;  $v_{\text{bb}} = 580$  nm/s (29),  $D = 0.083 \pm 0.001 \mu\text{m}^2 \text{s}^{-1}$  (10); for the protofilament model we assume  $b = 8$  nm and  $n = 13$  to be typical.

Table 1 in Franck et al. (12) (see Table S1 in the Supporting Material) lists velocities at  $f = 0.5$  pN and 2.0 pN. Using the velocity data we fit to obtain  $\delta = 14 \pm 1.4$  nm and  $v_{\text{ps}} = 55 \pm 9.3$  nm/s. As has already been mentioned the range  $\delta$  is intriguingly close to a tubulin axial repeat length (8 nm or 1.5 times this, due to helicity). A simple picture might be of a sleeve that suppresses depolymerization while it sits over the next intact tubulin dimers in the PFs that are about to split. This supports the idea that the MT depolymerizes by first splitting in a linear fashion, perhaps along its seam, with the other PF pairs splitting apart somewhat behind this leading crack-like defect. Indeed, materials do typically split along linear cracks, where the elastic stresses are concentrated (45). In particular, splitting between random PFs would yield step sizes that, due to helicity, could be a small fraction of the tubulin size. It would be hard to physically motivate a range  $\delta$  that is more than ten times the incremental depolymerization step size. Why would such a depolymerization process, involving little or no motion of PFs more than 1 nm from the last fully polymerized section of MT, be highly sensitive to the presence of a Dam1 ring more than 10 nm distant? We therefore consider our

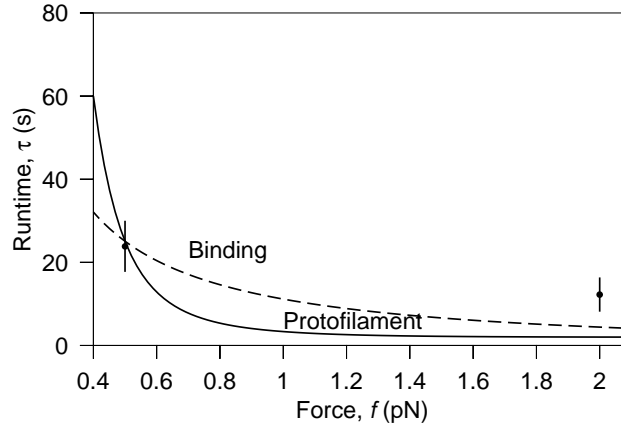


Figure 5: Runtime of the protofilament and binding models. The runtime  $\tau$  of each model is calculated using the parameters fitted as described in the Results section. Although, it may seem that distinguishing the models by varying force is possible due to the difference between their predicted behaviour, as shown here, the difference is close to experimental error ( $\pm 6.15$  s) and both models present similar functional form. Only two data points with sufficient statistics were available to perform this fitting (12) making it difficult to draw any conclusions from this approach. The fit provides values for  $\Delta G_{\text{Dam1}}$  for the binding model and  $k_{\text{break}}$  for the protofilament model.

estimate of the characteristic range  $\delta$  for the burnt-bridges reaction (within which the Dam1 ring occludes unzipping) to be quite reasonable.

Combining the available data for velocity and detachment frequency we find, on average,  $\tau = 23.9$  s and 12.2 s, for  $f = 0.5$  pN and 2.0 pN respectively. To fit the binding model for  $\tau$  we choose  $\varepsilon = 1$  nm, as a reasonable distance over which an attraction might act, and find  $\Delta G_{\text{Dam1}} = 15 \pm 0.26 k_B T$ . Independently, we fit  $k_{\text{break}}$  for the protofilament model and find  $k_{\text{break}} = 7.1 \pm 0.63 \text{ s}^{-1}$ . Fitting these parameters to just two data points does not provide strong evidence for these particular values. However, uncertainty in the exact parameter values should not detract from the main value of this work; to provide a model that explains the Dam1 force sensitivity and to distinguish between binding and protofilament models. Comparison of the fit with data is shown in Fig. 5.

### Variation of intrinsic depolymerization velocity

The protofilament model exhibits the most sensitivity to the intrinsic (bare) MT depolymerization velocity  $v_{\text{bb}}$ , as is shown in Fig. 6 A. For the protofilament model,  $\tau$  is strongly dependent on  $\langle t_{\text{unzip}} \rangle$  and consequently  $v_{\text{bb}}$ . The binding model, on the other hand, is only weakly dependent on  $v$  (see *Supporting Material*), and on this range of  $v_{\text{bb}}$  we can assume that depolymerization is quasistatically slow with respect to ring diffusion. The result can be understood physically by realising that as  $v_{\text{bb}}$  increases, the rate of PF unzipping  $k_{\text{unzip}}$  also increases, while  $k_{\text{break}}$  remains constant making it less likely that the PFs will break off sufficiently quickly to release the ring.

An experimental test that might be able to distinguish which model operates could be achieved, e.g., by addition of a depolymerization inducing agent, such as  $\text{Ca}^{2+}$  or XMCAK1.

### Changing of diffusion coefficient

The diffusion constant  $D$  of the ring is determined by the ring's dimensions and the roughness of the binding energy landscape along the MT, rather than the magnitude of the binding energy itself. A more rough landscape reduces the mobility of the ring. Fig. 6 B shows the effect of the diffusion constant on the runtime for both models. Only the binding model is sensitive to change in  $D$ , having reduced runtime with faster diffusion. This is because the increased mobility of the ring increases the chance it is able to scale the

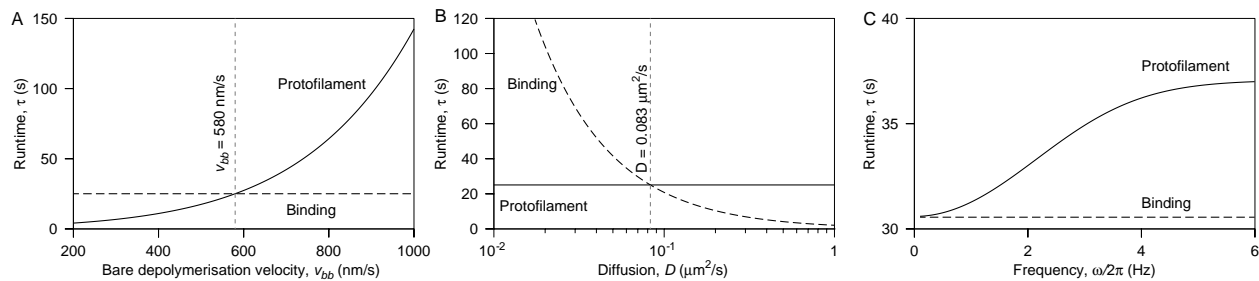


Figure 6: Model discrimination. The panels show variation of runtime  $\tau$  with (A) bare MT depolymerization velocity  $v_{bb}$ , (B) diffusion coefficient  $D$ , and (C) frequency of applied force  $\omega/2\pi$ , for both models under load  $f = 0.45$  pN, chosen because both models predict the same nominal  $\tau$  and  $v$  at this load (see Fig. 5). (A) The runtime  $\tau$  increases exponentially with  $v_{bb}$  for the protofilament model, whilst the binding model is insensitive. This is because the protofilament model directly depends on  $v$ , but the binding model does not. (B) Restricted diffusion suppresses detachment for the binding model because  $\tau$  is inversely related to  $D$ , due to the reduced impetus to escape the potential barrier. The protofilament model, on the other hand, is not affected by  $D$  since  $t_{\text{unzip}}$  is independent of  $D$ . Distinguishing between models will be easiest by experimental reduction of  $D$ , for example by attachment of a long polymer. (C) The binding model is sensitive only to the amplitudes  $f_0$  (here 0.1 pN) and  $f_1$  (here 0.43 pN), not the frequency  $\omega$ . The rate of detachment for the protofilament model instead strongly depends on the frequency: roughly speaking the ring is lost more quickly when the high force part of the cycle persists for long enough for the PFs to completely depolymerize in this time, i.e. when the period is long.

potential barrier constraining it to the MT.

Although it may be possible to alter  $D$  biochemically, for example by phosphorylation (30), it is difficult to do so independently of  $\Delta G_{\text{Dam1}}$ . Decreasing  $D$  may be better accomplished by attaching a long inert polymer to the complex to increase viscous drag.

### Effect of time-varying loading force

The runtime in the binding model is sensitive only to the instantaneous force provided  $2\pi/\omega \gg \lambda^2/D$ , see the low frequency portion of Fig. 6 C. If  $f_1 = 1$  pN then  $2\pi/\omega_{\text{max}}$  is on the order of 1 kHz. The runtime in the protofilament model is sensitive to the time over which changes in  $v$  persist. If the force is oscillating with a long period then the rate of detachment will be greater in the high force part of the cycle than if the period is short. This is because the Dam1 ring takes some time to detach if it needs to first wait for the PF curls to break, see Fig. 6 C. Sigmoidal increase of  $\tau$  would be a signature of a system that depends on a second time ( $1/k_{\text{break}}$ ), like the protofilament model; insensitivity of  $\tau$  to frequency would imply a binding-style coupling.

## Conclusion

Our results indicate a power stroke does contribute to the effective force generated during depolymerization but only becomes dominant at over 2 pN load. We show how a faster depolymerization mechanism must operate at lower loads and argue that the Dam1 ring suppresses depolymerization when it is close to the MT end.

We have shown that either of two rather different Dam1-MT coupling mechanisms might be operating under piconewton loads. Both models have comparable performance under load; their differences only become apparent under novel experimental conditions. Structural studies cannot resolve the question of which model operates *in vivo*. We suggest several methods for using runtime statistics to determine which class of model best describes the coupling of the Dam1 ring to depolymerizing MTs. Note that throughout this study we have assumed that depolymerization is sufficiently slow compared to ring diffusion that we can consider the distribution of the ring's position to be quasi-equilibrated. Over the range of parameters we have considered this assumption is valid to within 1% of the predicted velocity. The characteristic

range over which Dam1 inhibits depolymerization can be estimated by comparing our model with data. It is intriguingly close to the size of the microscopic (tubulin) repeat length of the MT. We have argued that this provides evidence that the MT is splitting, possibly along its seam, at the leading edge of the depolymerization front. In this case the PFs move outwards a similar distance along the MT from the last polymerized section.

It is important to note that the present work has neglected *in vivo* factors such as microtubule-associated proteins (MAPs) or kinases. However, some of these factors operate to increase or reduce the depolymerization rate of the microtubule, a parameter included in the model. We therefore expect the general results to remain largely applicable. Furthermore, we have assumed Dam1 to be present as a ring. Recent work (30, 31) has raised the possibility that Dam1 may operate as short oligomers or single complexes. If we can assume these oligomers interact with PFs in a comparable fashion as a ring would, our model would be indistinguishable for rings or oligomers. If not, our model may be of use to determine whether ring or oligomer is present based on e.g., differing diffusion constant.

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